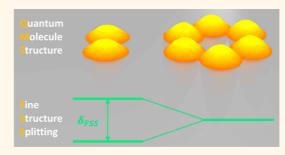


Exciton Fine-Structure Splitting in Self-Assembled Lateral InAs/GaAs Quantum-Dot Molecular Structures

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ABSTRACT Fine-structure splitting (FSS) of excitons in semiconductor nanostructures is a key parameter that has significant implications in photon entanglement and polarization conversion between electron spins and photons, relevant to quantum information technology and spintronics. Here, we investigate exciton FSS in self-organized lateral InAs/GaAs quantum-dot molecular structures (QMSs) including laterally aligned double quantum dots (DQDs), quantum-dot clusters (QCs), and quantum rings (QRs), by employing polarization-resolved microphotoluminescence (*µ*PL) spectroscopy. We find a clear trend in FSS between



the studied QMSs depending on their geometric arrangements, from a large FSS in the DQDs to a smaller FSS in the QCs and QRs. This trend is accompanied by a corresponding difference in the optical polarization directions of the excitons between these QMSs, namely, the bright-exciton lines are linearly polarized preferably along or perpendicular to the $[1\overline{10}]$ crystallographic axis in the DQDs that also defines the alignment direction of the two constituting QDs, whereas in the QCs and QRs, the polarization directions are randomly oriented. We attribute the observed trend in the FSS to a significant reduction of the asymmetry in the lateral confinement potential of the excitons in the QRs and QCs as compared with the DQDs, as a result of a compensation between the effects of lateral shape anisotropy and piezoelectric field. Our work demonstrates that FSS strongly depends on the geometric arrangements of the QMSs, which effectively tune the degree of the compensation effects and are capable of reducing FSS even in a strained QD system to a limit similar to strain-free QDs. This approach provides a pathway in obtaining high-symmetry quantum emitters desirable for realizing photon entanglement and spintronic devices based on such nanostructures, utilizing an uninterrupted epitaxial growth procedure without special requirements for lattice-matched materials combinations, specific substrate orientations, and nanolithography.

KEYWORDS: InAs/GaAs · quantum dot · quantum-dot molecular structure · fine structure splitting · microphotoluminescence · polarization

odern nanoscale engineering offers us a controllable way to build and design materials from nanoscale building-blocks.¹⁻⁴ Among many interesting research topics in the field of nanoscience and nanotechnology, semiconductor quantum dots (QDs) and more complex quantum-dot molecular structures (QMSs) are important keys for realizing novel photonic, spintronic and quantum information devices including nanoscale solid-state lighting,⁵ entangled photon sources for quantum teleportation, quantum cryptography and distributed quantum computation,⁶⁻⁸ and spin light-emitting devices and quantum information processing utilizing electron spins.^{9–11} This is largely motivated by

the expectations that QDs are efficient light emitters and possess long electron spin relaxation/coherence time.^{12,13} The photons emitted from the ODs are a result of excitonic recombination, which is associated with electron-heavy hole (e-hh) exchangecoupled pairs in most of the known semiconductor QD systems. It has now been well established that an anisotropic exchange interaction (AEI) can split the doublet of the bright neutral exciton states into two spin-mixed, linearly polarized |H> and |V> exciton states, commonly referred to as a fine-structure-splitting (FSS).¹⁴ In highly strained self-assembled InAs/(Al)GaAs QDs, FSS has been shown to be in the range of $40-1000 \,\mu \text{eV}$.¹⁵ This strong variation in FSS

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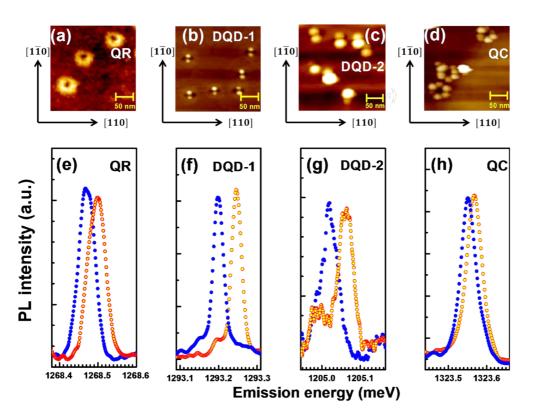


Figure 1. (a–d) AFM images of the studied QRs, laterally aligned DQDs and QCs. (e–h) Representative μ PL spectra obtained from single QR, DQD, and QC, when the axis of the linear polarization analyzer was aligned along the [110] (the solid symbols) and [110] (the open symbols) axes. The displayed spectra from the QRs and QCs are special cases when the polarization axes of the exciton states coincide with these crystallographic axes.

was suggested to be governed by effects of strainand shape-induced wave function asymmetry of the excitons.^{15,16} The appearance of FSS becomes a major problem in many applications exploiting QDs. For example, FSS results in two distinguishable paths of exciton-biexciton cascade decays of different polarizations, inhibiting entanglement of the resulting photon pairs necessary for quantum information applications. Moreover, the presence of FSS signifies a reduced symmetry of the zero-dimensional quantum emitters where AEI induces strong electron spin relaxation and mixing of exciton spin states, undesirable for future applications in spintronics, spin-photonics and spin-based quantum information processing. During the past decade, we have witnessed intense research efforts and achievements in suppressing FSS by nanoscale engineering, e.g., droplet epitaxy to achieve lattice matched QD systems,¹⁷ shape-controlled growth by nanohole filling,¹⁸ and fabrication of QDs on a highsymmetry crystallographic plane (111)A¹⁹ or (111)B substrate.²⁰

In this work, we demonstrate a different approach, namely geometrical engineering of self-organized QMSs, where FSS even in a strained QD system such as InAs/GaAs QDs can be significantly reduced to a limit similar to that achieved in strain-free QDs. This approach is attractive as the growth procedure is rather simple, using an uninterrupted molecular beam epitaxy (MBE) process within a single growth run,^{21,22} and does not require lattice-matching, special substrate orientations, nanolithography and site-controlled growth that could often be technically challenging and costly.

RESULTS AND DISCUSSION

Typical atomic force microscopy (AFM) images of the studied QMSs are shown in Figure 1a-d. First, all QMSs in a given sample are rather uniform in their overall sizes as a whole though the sizes of individual QDs forming each QMS noticeably differ. The guantum rings (QRs) are slightly elongated along a (110) crystallographic axis in the (001) plane. For the double quantum dots (DQDs), the two dots constituting a DQD were preferably aligned along the [110] direction in the (001) plane, though each individual dot is slightly elongated along the orthogonal [110] direction, see Figure 2d for a close-up AFM image. Two samples of the DQDs, differing in QD sizes, are studied in this work to examine the size effect. In the QCs, each cluster typically consists of 5-7 dots that are rather symmetrically arranged around the center. There are, however, large variations in sizes and the directions of the elongation axes between individual QDs within each QC.

The zero-dimensional nature of the studied QMSs was confirmed by the observation of discrete sharp excitonic emission lines, as revealed in our microphotoluminescence (μ PL) studies of individual QR, DQD,

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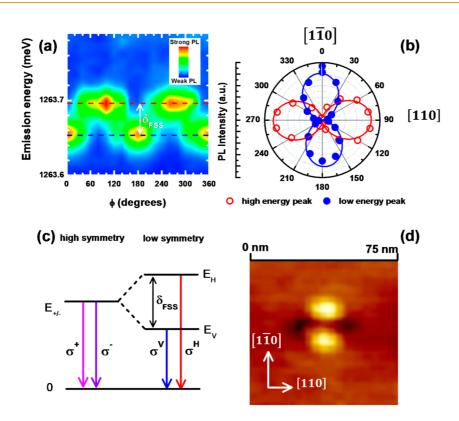


Figure 2. (a) 2D mapping of the bright ground states of the neutral exciton. (b) A polar plot of the integrated intensity of the lower-energy (the filled circles) and higher-energy (the open circles) exciton states. The solid lines are the simulated curves by eq 1. (c) Energy diagram representing the effect of AEI on the bright exciton states. (d) A close-up AFM image of a single DQD. All data were obtained from an individual DQD of the DQD-1 sample.

and QC shown in Figure 1e-h. The spectral line width of \sim 15–35 μ eV is in agreement with previous studies of OMSs.²³⁻²⁵ We note that this line width is much larger than what is expected from the lifetime broadening or the spectral resolution of our instrument (\sim 5 μ eV). We believe that the main reason for the observed inhomogeneous broadening of the PL line width is the random fluctuations of local potentials as a result of, e.g., fluctuating electric fields induced by charges at neighboring QDs and C-acceptors in the GaAs layers.²⁶ When we measure μ PL spectra in two orthogonal polarization directions, the doublet nature of the bright exciton ground states can be clearly resolved for each QMS. To determine the polarization axes of these two PL lines, we measured μ PL spectra as a function of the rotation angle ϕ of the linear polarization analyzer with respect to the [110] crystallographic axis. Figure 2a displays representative results, from an individual DQD as an example, which show a strong angular dependence in the relative PL intensities of the two excitonic lines, while their energy positions and thus the FSS remain constant for the given DQD as expected. A polar plot of the intensity of each PL line is shown in Figure 2b, obtained from a deconvolution analysis of each μ PL spectrum by two PL components of the fixed energies. The angular variation in the intensity of each PL line can be modeled by the following

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equation:

$$I = I_{\min} \sin^2(\phi - \theta) + I_{\max} \cos^2(\phi - \theta)$$
(1)

where $I_{max}(I_{min})$ defines the maximum (minimum) PL intensity for a given PL line. The phase θ is the angle at which the concerned PL line reaches I_{max} , which corresponds to the angle between the optical polarization axis and the $[1\overline{1}0]$ axis. An example of such simulations is given by the solid lines in Figure 2b, where the lower-energy line is found to be polarized along the $[1\overline{1}0]$ direction, whereas the higher-energy line is orthogonally polarized (*i.e.*, along the [110] axis). This polarization behavior is characteristic for an AEIinduced FSS of the bright neutral exciton states.¹⁴ As illustrated in Figure 2c, without AEI the two bright states of the neutral excitons are 2-fold degenerate with the eigenstates of $|+1\rangle$ and $|-1\rangle$ that are σ^+ and σ^- active, respectively. The electron spin is preserved in these hh exciton states, namely, electron spin-down in $|+1\rangle$ and electron spin-up in $|-1\rangle$. The action of a nonzero AEI reduces the symmetry (for example from D_{2d} to C_{2v}), which lifts the degeneracy of the doublet and splits it into two nondegenerate exciton sublevels denoted by H and V. Their eigenstates are $|H\rangle = (|+1\rangle +$ $|-1\rangle$)/ $\sqrt{2}$ and $|V\rangle = (|+1\rangle - |-1\rangle)/\sqrt{2}$ [ref 14], respectively, in which the electron spin states are strongly mixed. These two sublevels give rise to two linearly and orthogonally polarized PL lines, *i.e.*, σ^{H} for H and σ^{V} for V.

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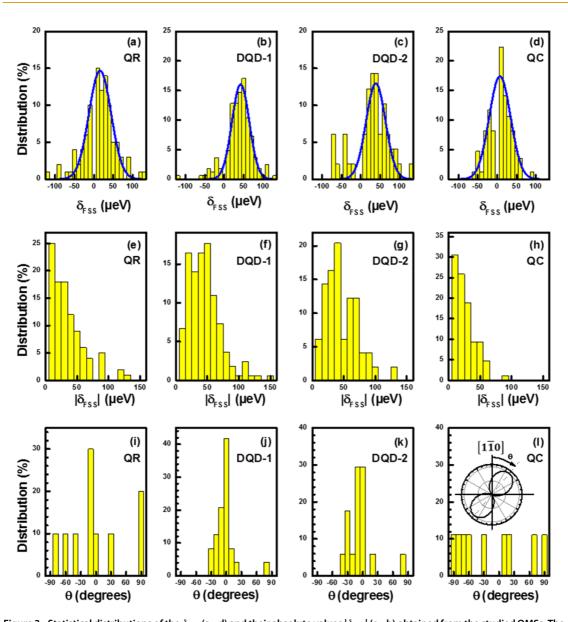


Figure 3. Statistical distributions of the $\delta_{FSS}(a-d)$ and their absolute values $|\delta_{FSS}|$ (e–h) obtained from the studied QMSs. The solid curves in (a–d) represent a Gaussian distribution function, as a guide to the eye. (i–l) Angular distributions of the polarization axis of the lower-lying exciton state from the studied QMSs, with respective to the [110] axis as defined by the sketch in (l). The total numbers of the individual QMSs used in the statistical distributions are 100 in (a) and (e), 164 in (b) and (f), 49 in (c) and (g), 85 in (d) and (h), 10 in (i), 24 in (j), 17 in (k), and 9 in (l).

The magnitude of the FSS (δ_{FSS}) can be determined by the energy separation between the H and V sublevels, e.g., $|\delta_{FSS}| = |E_H - E_V| \approx 40 \,\mu\text{eV}$ for the DQD-1 shown in Figure 2a. The linear polarization directions of the H and V state are defined by the directions of exciton wave function asymmetry under the influence of a directional confining potential. In the case of the DQD shown in Figure 2a,b, the lower-lying state E_V is polarized along the [110] axis, whereas the higher-energy state E_H is polarized along the [110] axis as indicated in Figure 2d.

To obtain more detailed information on the average value of the FSS and the polarization axis of the exciton sublevels and to reliably compare them between the studied QMSs, we carried out a systematic and statistical study of the FSS in a large number of individual QMSs for each QMS ensemble. The measured distributions of the δ_{FSS} values and their absolute values $|\delta_{FSS}|$ in the studied QMSs are displayed in Figure 3a–d and Figure 3e–h, respectively. The data points with $\delta_{FSS} < 5 \,\mu\text{eV}$ are not shown in Figure 3a–h because the corresponding single PL lines can be contributed by both neutral excitons with $|\delta_{FSS}| < 5 \,\mu\text{eV}$ (thus unresolved due to the spectral resolution of our μ PL system) and charged excitons such as positive and negative trions that remain spin degenerate due to the Kramers theorem. Since these two contributions cannot be definitely separated experimentally, no reliable data points for the concerned neutral excitons can be obtained. However, one would

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TABLE 1. Average and Most Probable Values of $\delta_{\rm FSS}$ and $|\delta_{\rm FSS}|$ Determined from the Statistical Distributions in the Studied QMS Ensembles^a

	QR	DQD-1	DQD-2	QC
Average δ^{av}_{FSS}	12	38	28	7
Most probable δ^p_{FSS}	16	42	39	7
Average $ \delta_{FSS} ^{av}$	28	43	43	19
Most probable $ \delta_{FSS} ^p$	10	50	40	10

^{*a*} The average values are calculated by $\partial_{FSS}^{av} = (\Sigma_n \partial_{FSS}^{av} [\Sigma_n, \partial_{FSS}) [\Sigma_n, \partial_n] |\partial_{FSS}|^{av} = (\Sigma_m |\partial_{FSS}^{av} [\Sigma_n, \partial_n] [D_{FSS}]^{av}$, respectively, where $n_i (m_i)$ represents the number of the QMS with the FSS value of $\partial_{FSS}^{i} [(\partial_{FSS}^{i}])$. The most probable values of ∂_{FSS}^{av} and $|\partial_{FSS}|^{av}$ correspond to the peak positions in the Gaussian distributions shown in Figure3a–d and in the experimental data in Figure 3e–h, respectively. The predicted values around zero FSS from the Gaussian distribution of ∂_{FSS} are used to estimate the values given in the table.

expect that such data points should fall within the statistical distributions given in Figure 3a-h. Two prominent features can clearly be seen from the results shown in Figure 3a-h. First, the FSS in the DQDs is found to be noticeably larger than that obtained in the QRs and the QCs, both in δ_{FSS} and $|\delta_{FSS}|$ values as given in Table 1. Second, a sizable portion of the QRs and QCs appears to have a negative δ_{FSS} value, whereas the corresponding proportion is significantly less in the DQDs. The positive (negative) sign of $\delta_{\rm FSS}$ is assigned to a QMS when the lower-energy exciton line of the bright exciton doublet becomes more polarized along the $[1\overline{1}0]$ ([110]) axis. The situation of a negative δ_{FSS} arises when the polarization direction of the lower-lying exciton sublevel has undergone a rotation away from $[1\overline{1}0]$ by an angle θ larger than 45°, such that its projected intensity along the $[1\overline{1}0]$ axis becomes weaker than that along the [110]axis. The observed larger spread in the distribution of δ_{FSS} toward negative values is thus indicative of a strong deviation between individual OMSs within the ensemble in the optical polarization axis and thus the direction of the expansion of the exciton wave function.

This assertion is further supported by the statistical distribution in the angle θ between the polarization axis of the lower-energy exciton state and the $[1\overline{1}0]$ crystallographic axis, experimentally obtained from a detailed angular dependent study of the polarization with the aid of eq 1 in the same way as that shown in Figure 2b. The results from the QMSs studied in this work are summarized in Figure 3i-l. It is obvious that the polarization axis of the lower-energy exciton sublevel in the DQDs is largely pinned around $\theta = 0^{\circ}$. In other words, it is preferably aligned along the $[1\overline{1}0]$ axis. In sharp contrast, the polarization axis of the lower-energy exciton sublevel in the QRs and QCs is randomly oriented that signifies a lack of a preferred direction. This finding is consistent with the significantly larger spread in the sign of δ_{FSS} found in the QRs and QCs shown in Figure 3a–d, as δ_{FSS} changes its sign when the optical polarization axis undergoes a large rotation by more than 45°.

individual QDs are highly improbable to have the same electronic energies for carriers and excitons. Therefore, the excitons monitored under our experimental conditions, with low excitation density and at low measurement temperature, are most likely localized within a single QD. This is consistent with earlier studies of DQDs in which possible coupling between the two QDs in a DQD only occurs in the excited states of electrons.^{27,28} Consequently, the observed FSS of such a localized exciton is expected to be primarily determined by the anisotropy of the confinement potential in the QD where the exciton is localized. As the FSS induced purely by atomistic asymmetry is expected to be small (<10 μ eV),²⁹ the observed FSS should be mainly caused by shape- and strain-induced wave function asymmetry.^{15,16} Previous theoretical studies³⁰ and recent $k \cdot p$ calculations with a configuration interaction³¹ have shown that the lower-lying bright exciton state should be polarized along the direction parallel to the structural elongation axis of the QD, when only the effect of shape-induced asymmetry is taken into account. Our experimental results show an opposite correlation in the DQDs, namely, the lowerlying bright exciton state is in fact polarized along the $[1\overline{1}0]$ axis that is *perpendicular* to the elongation axis (i.e., [110]) of each of the two QDs constituting the DQD as shown in Figure 2b,d. This finding thus rules out the effect of the shape-induced asymmetry of individual QDs as being the governing factor for the sizable FSS observed in the DQDs. This conclusion is further supported by our experimental results from the statistical distributions of the lateral aspect ratio of individual QDs in the studied DQDs, QRs and QCs, shown in Figure 4. Unlike the trend found for the FSS with the geometric arrangement of the QMSs, no correlation can be found between the aspect ratio and the FSS. Furthermore, the DQD-1 sample with the smaller sizes and the larger lateral shape anisotropy of individual QDs than those in the DQD-2 sample should exhibit larger FSS¹⁵ if only the shape anisotropy is considered. This is inconsistent with our experimental finding that the FSS is in fact rather similar between the DQD-1 and DQD-2. These results show that factors other than the QD shape anisotropy must play a key role in determining the overall confinement potential of the excitons.

In principle, the observed FSS could originate either

from the QMS as a whole or from individual QDs that

make up the QMS. Judging from the different sizes

of the individual QDs found within each DQD and QC from the AFM images shown in Figure 1, these

Previous studies in strain-driven self-assembled QDs have suggested the strain effect as a dominant factor for the observed strong (>20 μ eV) and directional FSS.^{15,16} If strain-induced piezoelectricity (PZ) is included, the expansion of the exciton wave function (thus the direction of the polarization axis) can be either along the [110] or [110] direction, depending

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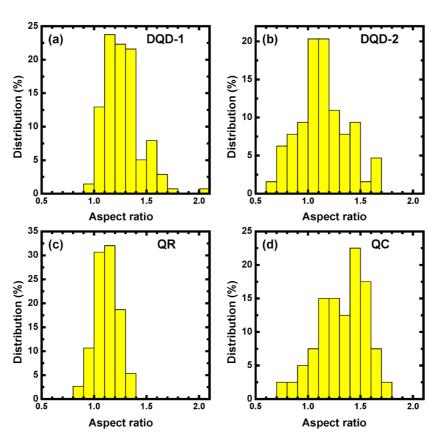


Figure 4. Statistical distributions of the lateral aspect ratios (long axis length/short axis length) obtained from the studied QMSs.

on the relative importance of linear and guadratic terms of the PZ as well as the correlation between electron and hole wave functions which are pointed orthogonally to each other under the action of the PZ.^{32–34} The rotation of the polarization axis by 90° away from the elongation direction of the QDs, as seen in the DQDs, provides evidence for an important role of a directional PZ field in inducing FSS. We therefore attribute the observed FSS to a combined effect of shape anisotropy and strain-induced piezoelectric field on the symmetry of the confinement potential of the excitons. These two effects compensate each other, and the resulting FSS thus depends on the degree of the compensation. In the case of the DQDs, the effect of the piezoelectric field substantially overcompensates the effect of the shape anisotropy such that the effective confinement potential is elongated along the [110] crystallographic axis despite the fact that the physical shape of each QD of the pair is actually elongated along the [110] axis. This leads to a sizable FSS and the [110]-oriented polarization axis of the lower lying bright exciton state. Since the degree of the shape anisotropy is similar among all studied QMSs, reflected by their similar lateral aspect ratios shown in Figure 4, a smaller FSS observed in the QRs and QCs as compared with that in the DQDs can then be explained by a reduction in the directional piezoelectric field so that its compensation to the effect

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induced by the shape anisotropy is more complete. The reduction in the piezoelectric field is related to a weak and less directional strain field experienced by each QR and each QD in the QCs, which is also responsible for the overall higher symmetry of these QMSs since their formation is strain driven. As the elongation direction of an individual QD in the QC is largely random and probably the piezoelectric field is autocorrelated with the OD shape anisotropy (thus strain field induced by In incorporation), the polarization axes of the bright exciton states also become randomly oriented in agreement with our experimental results. The suggested compensation between the effects induced by the shape anisotropy and the piezoelectric field is also consistent with our experimental finding that the FSS is nearly spectrally independent. This is because it has been shown that FSS induced by the same degree of the shape anisotropy should increase with increasing photon energy (i.e., for smaller QDs), whereas the piezoelectric field does the opposite,¹⁵ resulting in a cancellation in the spectral dependence of the FSS between the two effects. As to possible sources for the remaining small and nonvanishing FSS, besides an incomplete compensation between the effects of the shape anisotropy and the piezoelectric field, intrinsic effects such as interfacial symmetry lowering and its enhancement by atomistic elasticity,³⁵ atomic-scale alloy randomness³⁶

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and fluctuation of charged defects in the environment nearby the QMSs could also lead to a small value of the FFS ($\delta_{\rm FSS}$ < 10 $\mu {\rm eV}$) that occurs even in strain-free QD systems.

Our suggested model of the combined effect of the shape- and strain-induced anisotropy of the exciton confinement potential could also provide a possible explanation for the large FSS often reported in isolated single QDs grown in the strain-driven Stranski–Krastanov (SK) mode, *i.e.*, 40–1000 µeV.¹⁵ In the case of such isolated QDs, the shape elongation is usually found to be along the [110] crystallographic axis that coincides with the piezoelectric field. In sharp contrast to the compensation between the effects of the shape anisotropy and the piezoelectric field in our QMSs, these two effects are additive in the case of these isolated QDs. This leads to a large FSS reported in the isolated QDs, even larger than the DQDs studied in our work as one could expect.

The attractiveness and feasibility of the general idea in our approach of suppressing FSS by exploiting lateral QMSs is further bolstered by the success of QMS growth by a variety of growth methods besides the partial-capping-regrowth process employed in this work, such as the technique combining MBE and in situ etching,37-39 droplet epitaxy,40 partial capping and annealing,^{41,42} a hybrid growth approach utilizing both droplet homoepitaxy and SK growth on self-assembled nanomound templates,⁴³ etc.

CONCLUSIONS

In summary, our work demonstrated a possible pathway for suppressing FSS by nanoscale engineering exploiting the self-organized QMSs. We directly tackled the problem of strain-induced FSS by exploring controlled epitaxial growth conditions that lead to the formation of selective QMSs with a significantly reduced directional strain effect, without involving complicated growth and patterning procedures. More specifically, we started with QDs that could be used as the seeds to fabricate the QRs of a rather uniform size. These QRs possess a rather weak FSS, represented by an average and most probable absolute values of $|\delta_{FSS}|^{av}$ \sim 28 μ eV and $|\delta_{FSS}|^{p}$ \sim 10 μ eV, respectively. This FSS can be either enhanced or suppressed depending on the choices of parameters during the subsequent growth, which transforms the QRs into the DQDs or the QCs. The formation of the DQDs by the straindriven partial-capping-regrowth method resulted in a larger FSS with $|\delta_{\text{FSS}}|^{\text{av}} \sim 43 \,\mu\text{eV}$ and $|\delta_{\text{FSS}}|^{p} \sim 45 \,\mu\text{eV}$. The partial-capping-regrowth of the QRs at a higher temperature, on the other hand, led to the formation of the QCs with a small FSS with $|\delta_{FSS}|^{av}$ \sim 19 μ eV and $|\delta_{\text{FSS}}|^p \sim 10 \,\mu\text{eV}$, the smallest FSS in all studied QMSs. This small FSS is equivalent to the values found for strainfree QDs such as GaAs/AlGaAs QDs reported in the literature, but can now be achieved in a nominally strained QD system based on InAs/GaAs QDs in a selforganized fashion. The physical mechanism for the observed trend in the FSS among the QMSs is discussed in terms of the compensation between the effects of the shape anisotropy and the piezoelectric field on the resulting confinement potential of the monitored excitons localized within individual QDs constituting the QMSs. Our work has thus demonstrated an attractive alternative approach for producing high-symmetry guantum emitters via geometric engineering of QMSs, desirable for future applications of entangled photons and electron/ exciton spins in guantum information processing/ communications and spintronics/spin-photonics.

METHODS

The studied samples are a set of self-organized InAs/GaAs QMSs grown in the SK mode by gas-source MBE on a (001) semiinsulating GaAs substrate. They include QRs, lateral-aligned DQDs, and QCs, formed during the partial-capping-regrowth process of self-assembled InAs QDs under different growth conditions. The QRs were formed by partially capping of InAs QDs by 6 monolayers (MLs) of GaAs at a substrate temperature of 470 °C under As₂ overpressure. By depositing an additional 0.6 ML of InAs on the QRs, InAs DQDs were formed. However, if the substrates were heated to 520 °C in the deposition step for the 0.6 ML of InAs, the QRs were transformed to QCs. All the samples are nominally undoped. For AFM studies, another laver of the same QMSs was regrown on the surface of each QMS sample. A detailed description of the QMS samples and the growth procedures can be found in refs 21 and 22. We note that GaAs capping could in principle lead to structural changes of buried QMSs, as compared with the QMSs on the surface imaged by AFM. The extent of the changes is expected to critically depend on details in methods, procedures and conditions used during the growth, which affect the surface chemical potential, the migration length and anisotropy of In adatoms, and In-Ga intermixing.^{27,42,44,45} It is extremely difficult to experimentally access and determine the exact shapes of

the buried QMSs, however, as any invasive approach can alter QMS structures by itself. AFM is at the moment the best method available for the purpose and has been used in nearly all studies reported in the literature and also in our work. Considering the relatively low temperatures and the arsenic species of As₂ used during the growth of our QMSs, of which both should lead to a shorter In migration length, we believe that the overall lateral arrangements of the buried QMSs are expected to be similar to their counterparts on the surface imaged by AFM.

The FSS of individual QMSs was studied by means of polarization-resolved μ PL spectroscopy performed at 5K. μ PL spectra were measured with a confocal system (Horiba-Jobin HR800) in the backscattering geometry. The continue-wave excitation source was either a wavelength-tunable Ti-sapphire laser (700-1000 nm) or fixed-wavelength solid-state semiconductor diode lasers, providing nonresonant optical excitation of the QMSs with photon energies above the GaAs bandgap. The excitation spot was \sim 1 μ m in diameter. The optical excitation power was kept low to avoid contributions from excited states and multiexciton complexes. The PL signal was dispersed by a high-resolution spectrometer, with a spectral resolution of about 5 μ eV, and detected by a charge-coupled-device (CCD) detector. For optical polarization studies, a linearpolarization analyzer, consisting of a rotatable half-wave-plate



in conjunction with a fixed linear polarizer, was placed in the detection path before the entrance slit of the spectrometer. Due to weak PL signals, a long accumulation time on the order of seconds was employed for acquisition of each PL spectrum. This is about 10⁹ times the exciton lifetime, and as a result, the measured PL spectrum was a statistical average over a large number of exciton decay events during which the monitored QD experienced significant local potential fluctuations. We should note that the potential fluctuations induce the PL line broadening mainly through a slight shift of the exciton energy. FSS measured from the PL peak positions, corresponding to the most probable local potential, is expected to be largely unaffected. Therefore, the impact of the PL line broadening on our analysis of FSS in the QMSs can be neglected.

Conflict of Interest: The authors declare no competing financial interest.

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